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# Nitrogen-doped carbon nanotubes with tunable structure and high yield produced by ultrasonic spray pyrolysis

Jian Liu, Yong Zhang, Mihnea Ioan Ionescu, Ruying Li, Xueliang Sun\*

Department of Mechanical and Materials Engineering, University of Western Ontario, London, ON, Canada N6A 5B9

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## ABSTRACT

Nitrogen-doped carbon nanotubes ( $CN_x$ ) were prepared by ultrasonic spray pyrolysis from mixtures of imidazole and acetonitrile. Imidazole, as an additive, was used to control the structure and nitrogen doping in  $CN_x$  by adjusting its concentration in the mixtures. Scanning electron microscopy observation showed that the addition of imidazole increased the nanotube growth rate and yield, while decreased the nanotube diameter. Transmission electron microscopy study indicated that the addition of imidazole promoted the formation of a dense bamboo-like structure in  $CN_x$ . X-ray photoelectron spectroscopy analysis demonstrated that the nitrogen content varied from 3.2 to 5.2 at.% in  $CN_x$  obtained with different imidazole concentrations. Raman spectra study showed that the intensity ratio of D to G bands gradually increased, while that of 2D to G bands decreased, due to increasing imidazole concentration. The yield of  $CN_x$  made from mixtures of imidazole and acetonitrile can reach 192 mg in 24 min, which is 15 times that of  $CN_x$  prepared from only acetonitrile. The aligned  $CN_x$ , with controlled nitrogen doping, tunable structure and high yield, may find applications in developing non-noble catalysts and novel catalyst supports for fuel cells.

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## 1. Introduction

Over the last decade, carbon nanotubes (CNTs) [1] have been widely studied due to their unique properties and potential applications in various fields, such as electronics, optics and energy conversion [2,3]. Depending on the requirement of different applications, the properties of CNTs can be modified through different functionalization methods, of which doping foreign elements is an effective and frequently used approach [4]. Due to its similar atomic size with carbon, nitrogen (N) is widely studied as a doping element in CNTs. It is revealed that the incorporation of N could change the nanotube structure [5], chemical reactivity [6], electrical conductivity [7,8], and mechanical properties [9] of CNTs. Due to the tunability of these properties, nitrogen-doped carbon nanotubes ( $CN_x$ ) have shown promising application potentials in various fields. For example, Dai and co-workers [10] have reported that aligned  $CN_x$  could provide high electrocatalytic activity for oxygen reduction reactions (ORR) in fuel cells as a metal-free electrode. It has been found that the field emission property of CNTs could not only be improved by N doping [11], but also be tailored by controlling the graphitic/pyridinic N substitution [12]. In addition, uniform and dense Pt nanoparticles have been achieved by

using  $CN_x$  as substrates [13,14]. The  $CN_x$  supported Pt nanoparticles exhibit greatly improved stability [13] and electrocatalytic activity [14] for ORR in proton-exchange membrane fuel cells, compared with Pt nanoparticles supported on regular CNTs.

Recently, the doping techniques, especially in a controllable manner, have been widely studied. Various synthesis methods have been developed to produce  $CN_x$ , including arc discharge [15], laser ablation [16], and chemical vapor deposition (CVD) [17]. The CVD method has been proven to be more controllable and more cost effective than the others. However, the practical application of  $CN_x$  requires the production of fairly large amounts of high-quality  $CN_x$  at a commercially viable price. Among various CVD based techniques, spray pyrolysis and aerosol assisted CVD (AACVD) seem to be suitable methods for industrial scale production of  $CN_x$ . For example, both methods allow continuous injection of carbon and catalyst precursors, and require no additional catalyst preparation step [18,19]. Using these two methods, it is possible to produce  $CN_x$  without any flammable or corrosive gases such as hydrogen and ammonia [18]. Looking at these two methods, one can find that spray pyrolysis can spray solution directly into the synthesis system at controlled flow rates [18], whereas AACVD can generate solution droplets with controlled sizes [19]. Nevertheless, these desirable features have not been realized in one setup till now. Herein, by combining the advantages of spray pyrolysis and AACVD, we designed an improved spray pyrolysis method, called *ultrasonic spray pyrolysis*. In ultra-

\* Corresponding author. Tel.: +1 519 661 2111x87759; fax: +1 519 661 3020.  
E-mail address: [xsun@eng.uwo.ca](mailto:xsun@eng.uwo.ca) (X. Sun).

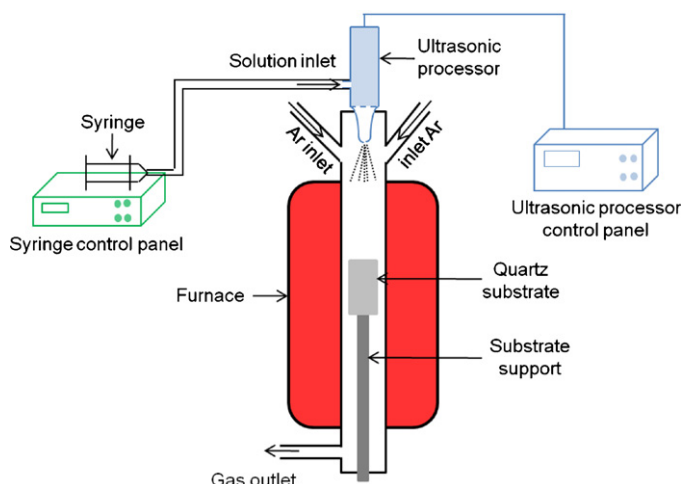


Fig. 1. Schematic illustration of ultrasonic spray pyrolysis.

sonic spray pyrolysis, an ultrasonic processor is used to generate fine solution droplets, which are directly sprayed into the synthesis system at controlled flow rates. Moreover, ultrasonic spray pyrolysis also features several other advantages including: (1) no need for preformed substrates; (2) possible use of mixtures of solid and liquid precursors; (3) easy scaling into an industrial scale process.

Besides the synthesis method, the controllable doping of N in CNTs also relies on the choice of precursor, catalyst, reaction tem-

perature, reaction time and gas flow rate [20–24]. Recently, some interesting progress has been made in controlling the morphology and structure of  $CN_x$  by using different liquid N precursors [21–23]. For example, a CNT array doped with a gradient of N concentration was achieved by gradually increasing the pyridine concentration in xylene [21]. Koós et al. reported that  $CN_x$  with different structures were prepared by using the mixture of benzylamine and toluene [22,23]. It was found that bamboo shaped nanotubes increased with the increase of benzylamine concentration, but the length and diameter of the nanotubes decreased dramatically [22,23]. In those studies, controlled doping of N in CNTs was realized by adjusting the concentration of the N-containing precursor in the N-free precursor. However, the involvement of the N-free precursor inevitably leads to decreased N content in  $CN_x$  [22], which is not desirable for the practical applications of  $CN_x$ . Moreover, the addition of N-containing precursor in N-free precursor significantly suppresses the growth of  $CN_x$  and results in a dramatically decreased yield [23,24], which makes it difficult to produce  $CN_x$  on a large scale.

In this work,  $CN_x$  with controlled N doping, tunable structure and high yield have been prepared by ultrasonic spray pyrolysis. The solution used for the synthesis of  $CN_x$  consists of a liquid N precursor (acetonitrile) and a solid N precursor (imidazole), acting as the solvent and the solute respectively. The use of this kind of solution avoids the decrease of N content in  $CN_x$ , because both components are N-containing precursors. And it is reported that imidazole can produce  $CN_x$  with an N content as high as 25.7 at.% [25]. Therefore, it is expected that the addition of imidazole in acetonitrile will increase the N doping in  $CN_x$ .

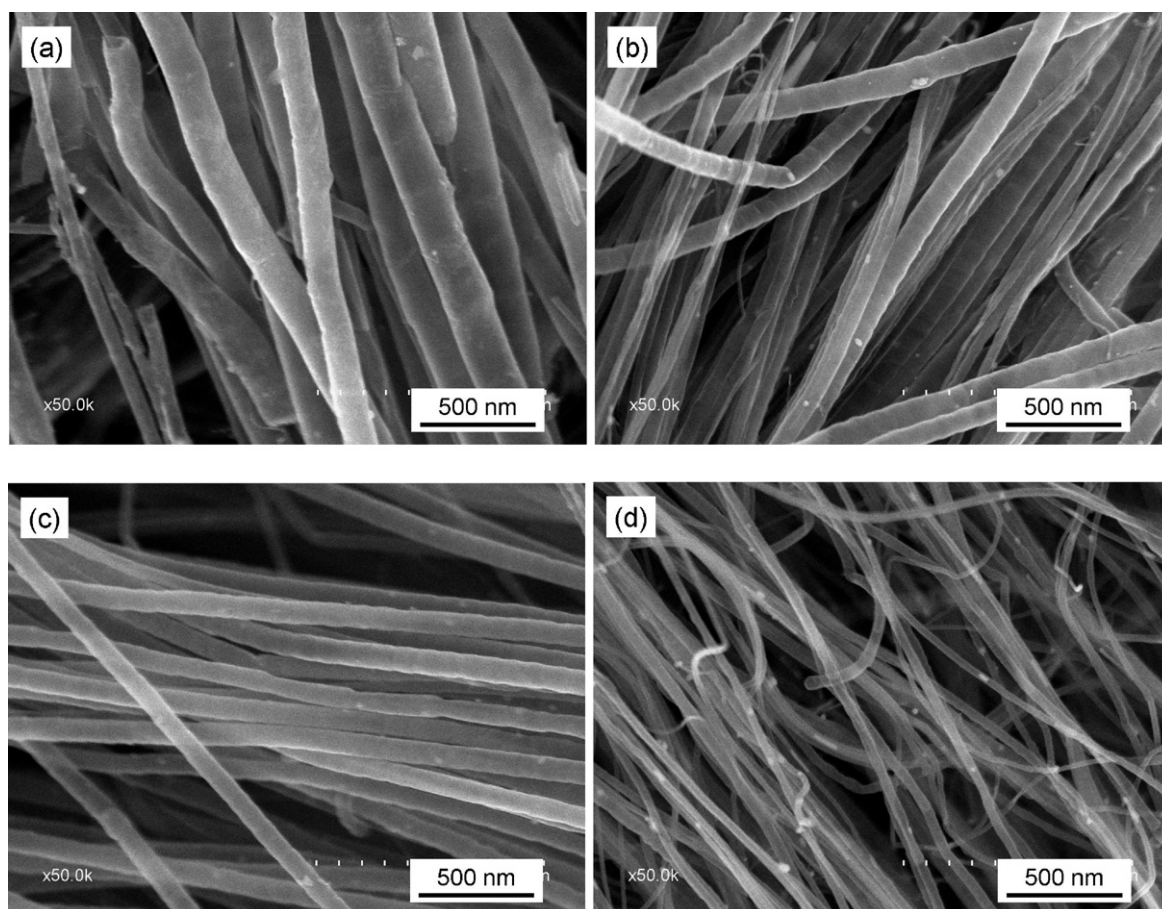
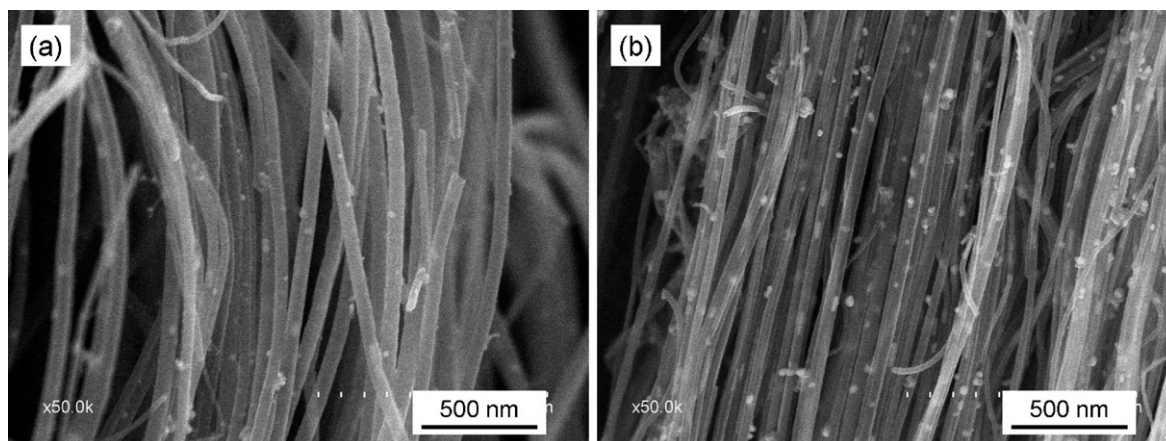


Fig. 2. SEM images of  $CN_x$  produced from solutions with different imidazole concentrations: (a) 0 mg/ml; (b) 50 mg/ml; (c) 100 mg/ml; (d) 200 mg/ml. (Solution injection rate 0.25 ml/min; amplitude 45%).

**Table 1**Variation of mean nanotube diameter, nanotube length, growth rate and yield of CN<sub>x</sub> produced from solutions with different imidazole concentrations.

Imidazole concentration (mg/ml)	Mean nanotube diameter (nm)	Mean nanotube length (μm)	Growth rate (μm/min)	Yield (mg/6ml)
0	89 ± 5	78 ± 8	3.3	12
50	70 ± 4	112 ± 12	4.7	24
100	66 ± 4	154 ± 9	6.4	60
200	54 ± 3	397 ± 18	16.5	192

**Fig. 3.** SEM images of CN<sub>x</sub> produced at different solution injection rates (a) 0.5 ml/min; (b) 0.75 ml/min. (Imidazole concentration: 200 mg/ml; amplitude 45%).

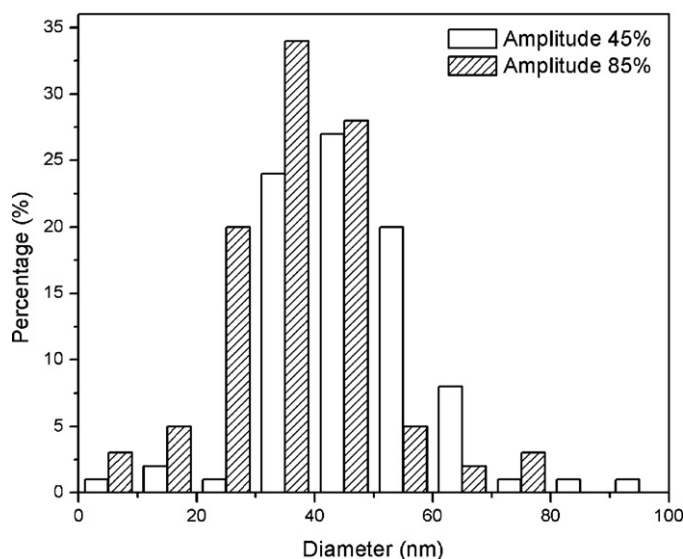
## 2. Experimental method

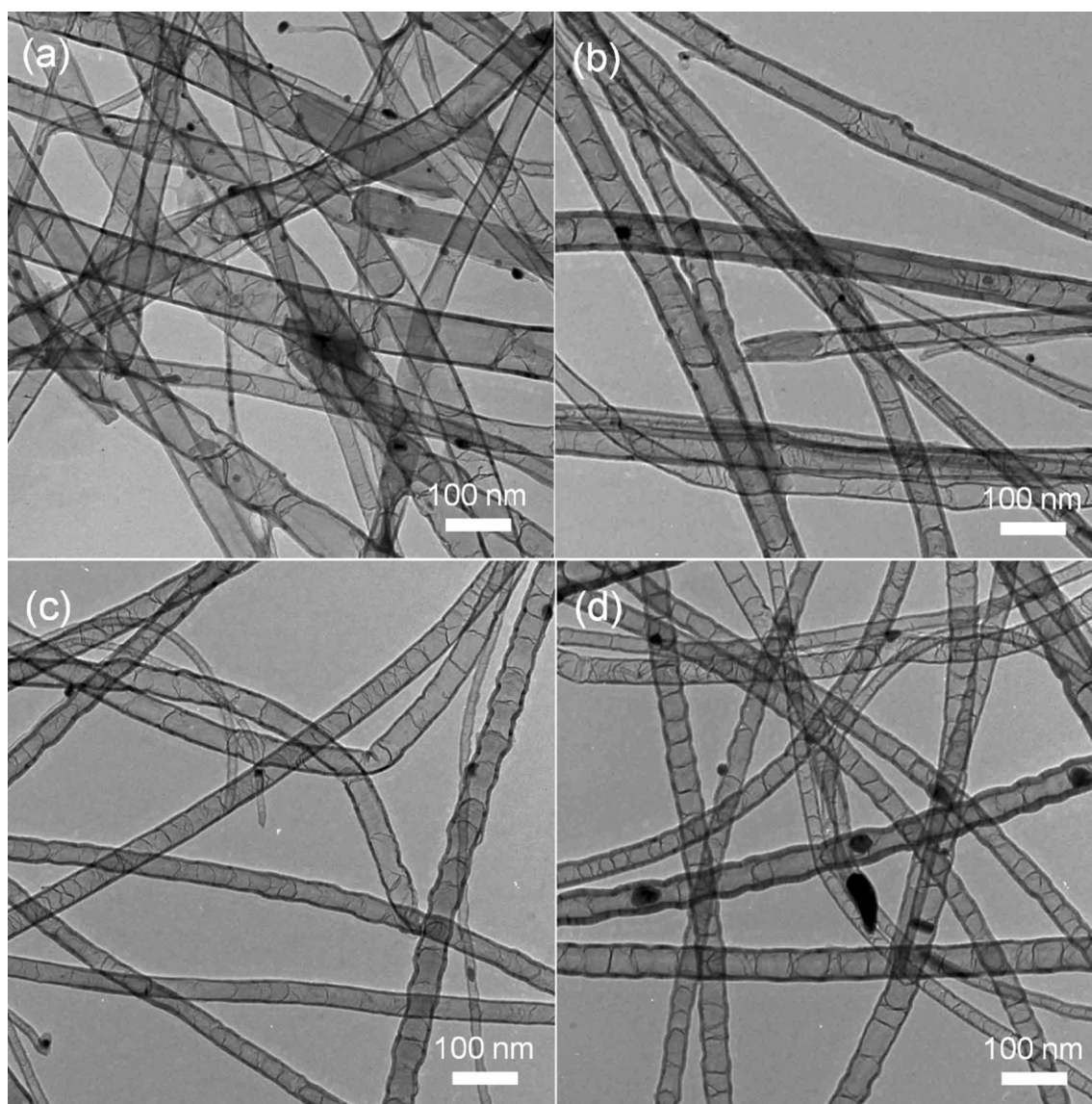
Fig. 1 shows the schematic illustration of ultrasonic spray pyrolysis. This setup mainly consists of three parts called the injection part, the ultrasonic part, and the deposition chamber. In the injection part, the solution stored in a syringe is injected into the ultrasonic part at a certain feeding rate, which can be precisely controlled on the syringe control panel. In the ultrasonic part, the solution fed by the injection part is atomized in an ultrasonic processor (VCX 130 PB, Sonics & Materials Inc.), and sprayed out as tiny droplets into the deposition chamber. Ultrasonic frequency of the solution can be controlled on the control panel of the ultrasonic processor. The deposition chamber consists of a vertical electrical furnace (50 cm length), a quartz tube (Ø 2.2 cm × 70 cm) with two gas inlets at the top and one gas outlet at the bottom, and a quartz plate (1 mm × 20 mm × 90 mm) attached on a substrate support as the substrate for the growth of CN<sub>x</sub>. Once sprayed out by the ultrasonic processor, the solution droplets will be carried by argon (Ar) gas into the center of the furnace, where the pyrolysis occurs.

In a typical process, different amounts (0.5, 1 and 2 g) of imidazole (C<sub>3</sub>H<sub>4</sub>N<sub>2</sub>, Alfa Aesar 99%) and 200 mg of ferrocene (Fe(C<sub>5</sub>H<sub>5</sub>)<sub>2</sub>, Aldrich 98%) were added into 10 ml of acetonitrile (CH<sub>3</sub>CN, 99.5+%), and the mixture was put into an ultrasonic cleaner for 10 min to obtain a homogeneous solution. The solution was then transferred into the syringe in the injection part. The quartz plate used as the substrate for the growth of CN<sub>x</sub> was placed in the center of the furnace, with the help of the substrate support. Before the furnace was heated, Ar (99.999% in purity) gas was introduced into the deposition chamber from both inlets for 20 min, and the Ar gas flow rate at each inlet was maintained at 150 sccm. Then the furnace was heated to 850 °C at a rate of 60 °C/min. Once the furnace reached the desirable temperature, 6 ml of the solution prepared as above was injected into the ultrasonic processor at different injection rates (0.25, 0.5 and 0.75 ml/min), and sprayed out as tiny droplets by the ultrasonic processor at different amplitudes (45% and 85%). Then those droplets were carried into the center of the furnace, where

the growth of CN<sub>x</sub> occurred. After 6 ml of the solution was injected, the furnace was turned off and the system cooled down to room temperature in the flowing Ar gas. CN<sub>x</sub> were collected from the quartz plate for future analysis.

The CN<sub>x</sub> were characterized by various analysis techniques including Hitachi S-4800 field-emission scanning electron microscopy (SEM) operated at 5 kV, Philips CM 10 transmission electron microscopy (TEM) operated at 80 kV, Kratos Axis Ultra Al (alpha) X-ray photoelectron spectroscopy (XPS) operated at 14 kV, and a HORIBA Scientific LabRAM HR Raman spectrometer operated with an incident laser beam at 532.03 nm.

**Fig. 4.** Diameter distribution of CN<sub>x</sub> produced at different ultrasonic amplitudes. (Imidazole concentration: 200 mg/ml; solution injection rate: 0.25 ml/min).



**Fig. 5.** TEM images of  $CN_x$  produced from solutions with different imidazole concentrations: (a) 0 mg/ml; (b) 50 mg/ml; (c) 100 mg/ml; (d) 200 mg/ml. (Solution injection rate 0.25 ml/min; amplitude 45%).

### 3. Results and discussion

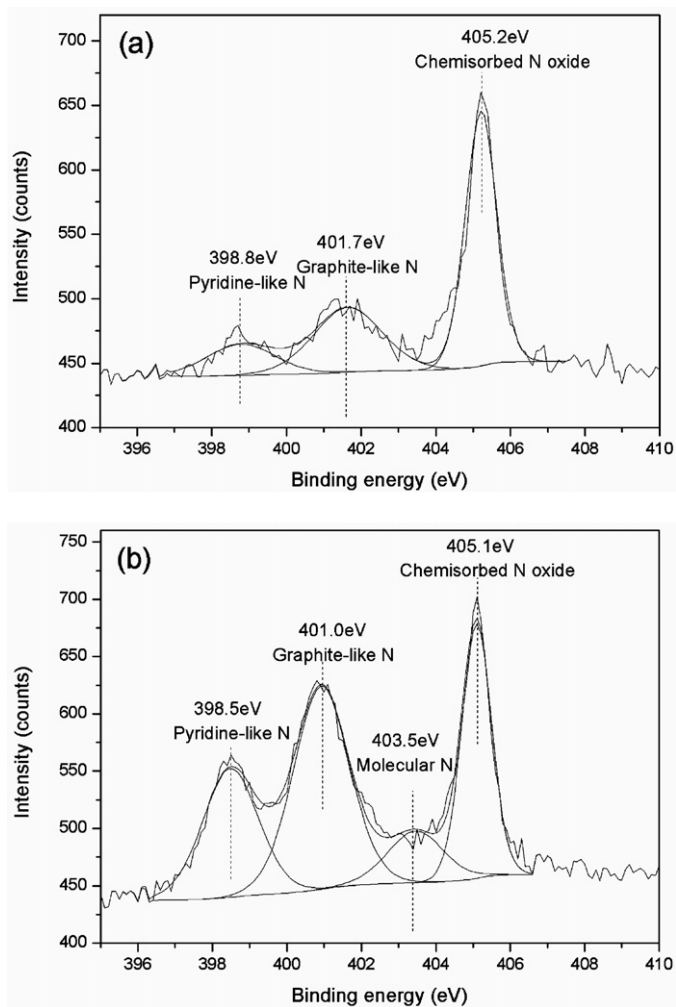
#### 3.1. Morphological investigation by SEM

##### 3.1.1. Effect of imidazole concentration

The effect of imidazole concentration on the growth of  $CN_x$  is studied in terms of the nanotube diameter, nanotube length, growth rate and yield. Fig. 2 shows the SEM images of  $CN_x$  prepared from solutions with different imidazole concentrations. It can be seen that the nanotube diameter of  $CN_x$  changes with the variation of the imidazole concentration. With the increase of the imidazole concentration, the nanotube diameter of  $CN_x$  gradually decreases. The diameters of at least 150 nanotubes were measured on TEM images of each sample. Based on these data, their mean diameters are calculated and compared in Table 1. In Table 1, it can be found that when the imidazole concentration increases from 0 to 50, 100 and 200 mg/ml, the mean diameter of  $CN_x$  gradually decreases from 89 to 70, 66 and 54 nm, respectively. It is noteworthy that a narrower distribution of diameters can be achieved by increasing imidazole concentration. Koós et al. [22,23] reported that a reduction of nanotube diameter was observed by increasing benzylamine

(N-containing precursor) concentration in toluene (N-free precursor). A theoretical calculation [26] shows that the reduced nanotube diameter is mainly due to the presence of N in  $CN_x$ , which prefers to stay on the tube edge and thus induces the closure of the nanotubes. This is in accordance with our result. As shown in Section 3.3, the addition of imidazole into acetonitrile leads to the incorporation of more N atoms into  $CN_x$ , which is accompanied by the reduction of nanotube diameters.

The nanotube length of each sample was measured, and the results are listed in Table 1. In Table 1, it can be seen that the nanotube length of  $CN_x$  is greatly increased with the addition of imidazole. Without imidazole, the nanotube length of  $CN_x$  produced in 24 min is measured to be  $\sim 78 \mu\text{m}$ . With 200 mg/ml imidazole, the length of  $CN_x$  is found to be  $\sim 397 \mu\text{m}$ . Thus, the growth rate of  $CN_x$  is increased from  $3.3 \mu\text{m}/\text{min}$  for  $CN_x$  prepared without imidazole to 4.7, 6.4, and  $16.5 \mu\text{m}/\text{min}$  for  $CN_x$  obtained with 50, 100, and 200 mg/ml imidazole, respectively. Consequently, the yield of  $CN_x$  is increased with the addition of imidazole, as seen in Table 1. The yield of  $CN_x$  was obtained from the product scratched off from the quartz plate. With 200 mg/ml imidazole, 192 mg high-quality  $CN_x$  can be easily produced in 24 min. In comparison, the yield of



**Fig. 6.** XPS N 1s spectrum of  $\text{CN}_x$  produced from solutions with different imidazole concentrations: (a) 0 mg/ml and (b) 200 mg/ml.

$\text{CN}_x$  prepared without imidazole in 24 min is only 12 mg, which is only one fifteenth of the former. Therefore, it can be concluded that the addition of imidazole in acetonitrile promotes the growth of nanotubes, and leads to an increased nanotube length and yield for  $\text{CN}_x$ . Because imidazole can generate high quantities of C–N fragments at 850 °C [25], the solution with imidazole can provide more C–N units than that without imidazole for the growth of the nanotubes. Thus it is reasonable that the addition of imidazole in acetonitrile promotes the growth of  $\text{CN}_x$  by providing more C–N sources. The trend of nanotube yield in our case is different from previous studies [23,24], in which the addition of an N containing precursor in an N-free precursor led to suppressed growth of  $\text{CN}_x$ . The difference is due to the different solvents used in the solution. The solvent used in our case is an N containing precursor (acetonitrile), while the solvents used in other studies are N-free precursors. It is suggested that pre-existing C–N bonds in the precursor play a key role for the incorporation of N atoms in  $\text{CN}_x$  [27]. In our case, both imidazole and acetonitrile contain C–N bonds. Therefore, it is reasonable that the doping of N atoms and growth of  $\text{CN}_x$  would be easier by using the mixture of imidazole and acetonitrile than using a mixture of one N containing precursor and one N-free precursor.

### 3.1.2. Effect of solution injection rate

The effect of solution injection rate on the growth of  $\text{CN}_x$  is studied by using a solution with imidazole concentration of 200 mg/ml,

and the result is shown in Fig. 3. In Fig. 3, it can be seen that  $\text{CN}_x$  prepared at solution injection rates of 0.5 and 0.75 ml/min show similarity in the nanotube diameter, which is slightly larger than that of  $\text{CN}_x$  prepared at 0.25 ml/min, as seen in Fig. 2(d). For  $\text{CN}_x$  prepared at a solution injection rate of 0.75 ml/min, there are many small catalyst particles attached to the surface of nanotubes. This is due to that only part of the catalyst can catalyze the growth of  $\text{CN}_x$  at the high solution injection rate. Then the excess catalyst will deposit on the surface of  $\text{CN}_x$ , forming small catalyst particles. Therefore, the solution injection rate mainly affects the quality of  $\text{CN}_x$ .

### 3.1.3. Effect of amplitude

The amplitude of the ultrasonic processor is the last parameter investigated in our study. A solution with 200 mg/ml imidazole was used to grow  $\text{CN}_x$  at amplitude 45% and 85%. The diameter distribution of each kind of  $\text{CN}_x$  is shown in Fig. 4. It can be seen that the nanotube diameter of  $\text{CN}_x$  tends to decrease from 30–60 to 20–50 nm, as amplitude increases from 45% to 85%. The nanotube diameter is usually related to the size of the solution droplet [19]. Higher amplitude of ultrasonic processor can generate solution droplets with smaller sizes, which lead to smaller nanotube diameters. Therefore, the amplitude has an influence on the distribution of the nanotube diameter for  $\text{CN}_x$  prepared by ultrasonic spray pyrolysis.

## 3.2. Structural investigation by TEM

The internal structures of  $\text{CN}_x$  prepared from solutions with different imidazole concentrations are investigated by TEM, and their typical TEM images are shown in Fig. 5. In Fig. 5(a), it can be seen that without imidazole, the produced  $\text{CN}_x$  have large nanotube diameters and thin nanotube walls. There exist a few interlinks in the internal nanotubes. With the addition of 50 mg/ml imidazole, the nanotube diameter of the prepared  $\text{CN}_x$  becomes smaller, and interlinks in the nanotubes become denser than that of  $\text{CN}_x$  synthesized without imidazole, as seen in Fig. 5(b). When the imidazole concentration is further increased to 100 mg/ml, the obtained  $\text{CN}_x$  show similar internal structure with those prepared with 50 mg/ml imidazole. But their nanotube diameters are slightly decreased. In Fig. 5(d), the nanotubes of  $\text{CN}_x$  produced with 200 mg/ml imidazole show stacked-cone structure with periodic compartment separation, which is usually called bamboo-like structure. Comparing Fig. 5(a–d), it can be found that the addition of imidazole leads to the formation of a dense bamboo-like structure in the internal nanotubes of  $\text{CN}_x$ . It is widely accepted that the formation of bamboo-like structure is caused by the presence of N in the graphitic network in  $\text{CN}_x$ , which induces curvature of the graphitic layer [28,29]. A higher content of N doping usually leads to a denser bamboo-like structure in the internal nanotubes of  $\text{CN}_x$  [22,23,28]. In our case, the added imidazole generates many C–N fragments during the synthesis process, which introduces more N atoms into  $\text{CN}_x$  (proved by XPS result in Section 3.3). The doping of more N atoms in  $\text{CN}_x$  eventually promotes the formation of the denser bamboo-like structure in  $\text{CN}_x$  by inducing curvature of the graphitic layer. The TEM results suggest that the inner structure of  $\text{CN}_x$  can be controlled by simply adjusting the imidazole concentration in the solution.

## 3.3. Content and bonding environment of N by XPS

The content and the bonding environment of N are important factors that determine the properties of  $\text{CN}_x$ . In order to identify the effect of imidazole on the content and the bonding environment of N doping in  $\text{CN}_x$ , XPS analysis was carried out on  $\text{CN}_x$  prepared without imidazole and with 200 mg/ml imidazole, and the results

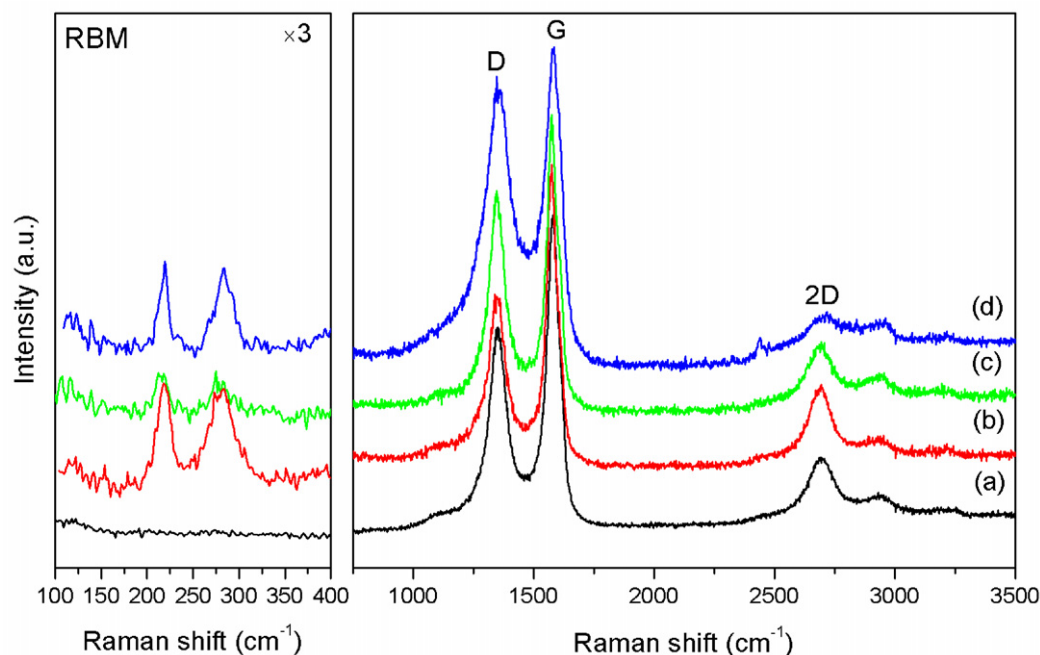


Fig. 7. Raman spectra of  $CN_x$  produced from solutions with different imidazole concentrations: (a) 0 mg/ml; (b) 50 mg/ml; (c) 100 mg/ml; (d) 200 mg/ml. (Solution injection rate 0.25 ml/min; amplitude 45%).

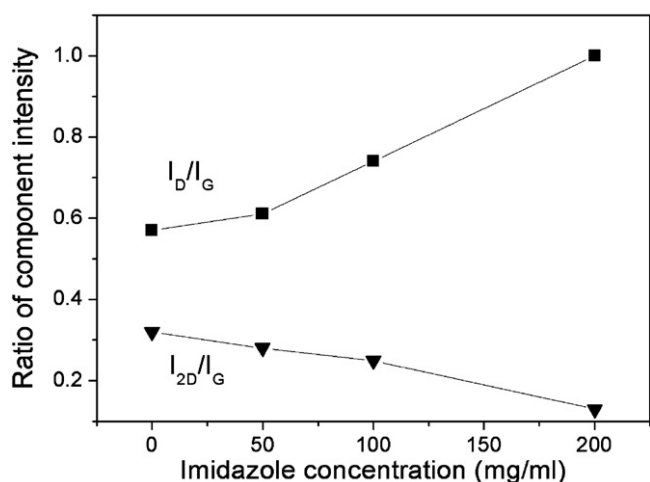


Fig. 8. Integrated intensity ratio of D to G bands ( $I_D/I_G$ ) and 2D and G band ( $I_{2D}/I_G$ ) as a function of the imidazole concentration.

are compared in Table 2 and Fig. 6. The N content, defined as atomic percent of  $N/(C+N)$ , is estimated by the area ratio between the N peak and the sum of N and C peaks. As seen in Table 2, the N content increases from 3.2 at.% for  $CN_x$  prepared without imidazole to 5.2 at.% for  $CN_x$  prepared with 200 mg/ml imidazole. The XPS N 1s spectra of both samples are fitted into three or four components located at 398.5–398.8, 401.0–401.7, 403.5 (if applicable) and 405.1–405.2 eV, corresponding to pyridine-like N, graphite-like N,

molecular N and chemisorbed N oxide, respectively [28], as seen in Fig. 6. The content of each kind of N is calculated and listed in Table 2. In Table 2, it can be seen that the contents of pyridine-like N and graphite-like N are obviously increased for  $CN_x$  produced with the addition of imidazole. The intensity ratio between pyridine-like N and graphite-like N ( $I_{pN}/I_{gN}$ ) is found to increase from 0.48 for  $CN_x$  without imidazole to 0.63 for  $CN_x$  with 200 mg/ml imidazole. It is noteworthy that molecular N appears in  $CN_x$  obtained with 200 mg/ml imidazole. Previous studies have indicated that molecular N can be encapsulated inside the tubes [29,30], or exist as intercalated form between the graphite layers during the growth of  $CN_x$  [31]. The content of chemisorbed N oxide is slightly decreased for  $CN_x$  obtained with 200 mg/ml imidazole, compared with those produced without imidazole. Therefore, the addition of imidazole not only increases the N content in  $CN_x$ , but also alters bonding environments of N in  $CN_x$ .

### 3.4. Crystallinity of $CN_x$ by Raman spectra

To obtain the information on the crystallinity of  $CN_x$ , Raman spectra were carried out on  $CN_x$  prepared with different imidazole concentrations, and their spectra are compared in Fig. 7. From Fig. 7, it can be seen that the Raman spectra show two main peaks around 1350 and 1590  $cm^{-1}$ , corresponding to the D-band and the G-band respectively. The D-band is originated from atomic displacement and disorder induced features caused by lattice defect, distortion or the finite particle size [32], while the G-band indicates the formation of well-graphitized carbon nanotubes [33]. A second-order peak located at  $\sim 2700 cm^{-1}$  is also observed and usually named

Table 2

Total N content and contents of different types of N in  $CN_x$  produced from solutions with imidazole concentration of 0 and 200 mg/ml.

Imidazole concentration (mg/ml)	N content (at.%)				
	Total	Pyridine-like N	Graphite-like N	Molecular N	Chemisorbed N oxide
0	3.2	0.47	0.97	–	1.76
200	5.2	1.28	2.03	0.51	1.36

2D-band, and the 2D-band is an overtone mode of the D-band [34]. Previous studies [23,28,34] have shown that N doping in  $CN_x$  usually increases the intensity of D-band while decreases the intensity of 2D-band, due to increased defect density in the nanotubes. In our case, the increase of imidazole concentration in acetonitrile leads to an increase of the intensity of D-band and a decrease of the intensity of 2D-band. The intensity ratios of D to G and 2D to G are calculated to quantitatively show this variation, as shown in Fig. 8. As the imidazole concentration increases from 0 to 50, 100 and 200 mg/ml, the D/G ratio gradually increases from 0.57 to 0.61, 0.74 and 1, respectively. Meanwhile, 2D/G ratio decreases from 0.32 to 0.28, 0.25 and 0.13. The Raman results indicate that the addition of imidazole induces an increase in the defects and disorder in  $CN_x$ . The reason could be attributed to the doping of more N atoms in  $CN_x$  with the addition of imidazole.

It is interesting to note that the Raman spectra of  $CN_x$  prepared with imidazole show radial breathing mode (RBM) between 180 and 280  $cm^{-1}$ , which is the fingerprint of single-wall carbon nanotubes (SWNTs) [35]. The corresponding diameter of SWNTs is estimated in the range of 0.9–1.1 nm, by using the empirical relation  $d$  (nm) =  $248/\nu$  ( $cm^{-1}$ ), where  $d$  is the SWNTs diameter and  $\nu$  is Raman shift [35,36]. The appearance of SWNTs in  $CN_x$  is partially due to the decrease of nanotube diameters with the addition of imidazole in the solution. Another possible reason is the secondary generation of nanotubes during the synthesis process [22,37]. During the synthesis process, the surface of the quartz plate was totally covered by nanotubes within a few minutes. Then incoming catalyst could not reach the quartz plate, and would deposit on the nanotube surface for the growth of  $CN_x$  [22,37]. These new catalyst particles grown on the nanotube surface were much smaller than those grown on the surface of the quartz plate [37], and this increase the possibility of the appearance of SWNTs. Moreover,  $CN_x$  with higher N content have more active nanotube surface, which would be easier for the deposition of catalyst particles. Thus,  $CN_x$  with a high N content have more chance for the appearance of SWNTs than those with a low N content. This would explain why that SWNTs did not appear in  $CN_x$  prepared without imidazole. Therefore, ultrasonic spray pyrolysis also shows the potential in the synthesis of N doped SWNTs. Large amounts of N doped SWNTs are expected when the catalyst concentration is decreased in the solution.

#### 4. Conclusions

Ultrasonic spray pyrolysis was developed to synthesize nitrogen-doped carbon nanotubes ( $CN_x$ ) from mixtures of acetonitrile and imidazole. A systematic study was carried out on the effect of solution injection rate, amplitude of the ultrasonic processor and imidazole concentration on the growth of  $CN_x$ . The results showed that the solution injection rate mainly affected the quality of  $CN_x$ , and the amplitude had an influence on the diameter distribution of  $CN_x$ . SEM and TEM studies indicated that the increase of imidazole concentration decreased the nanotube diameter, while increased the nanotube length and the density of bamboo-like structure in  $CN_x$ . XPS and Raman spectra analysis showed that the addition of imidazole promoted the N doping and introduced more defects in  $CN_x$ . These results suggest that by simply adjusting the imidazole concentration in acetonitrile, it is possible to control the nanotube diameter and length, internal structure, N content and defect density in  $CN_x$  produced by ultrasonic spray pyrolysis.

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